

Research Activities towards long-term stable SOFCs at Montanuniversitaet Leoben

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Outline

▪ Focus

- Long-term investigation ($t < 1000$ h) of oxygen exchange kinetics of SOFC cathodes in real atmospheres
 - $\text{H}_2\text{O-}$, CO_2 -, SO_2 -containing atmospheres
 - surface poisoning by Cr- and Si-compounds

▪ Methods

- dc conductivity relaxation measurements (k_{chem} , D_{chem} , σ_e)
- surface and bulk characterisation by XPS and TEM (EDXS, EELS)
- thermodynamic calculations

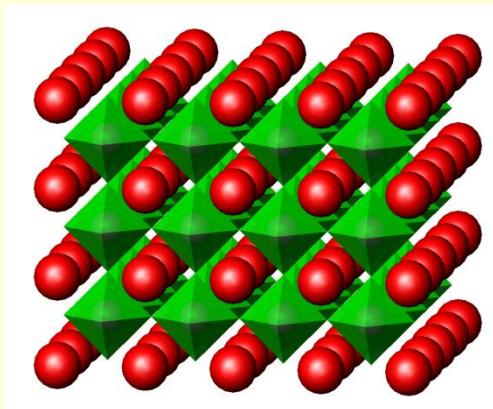
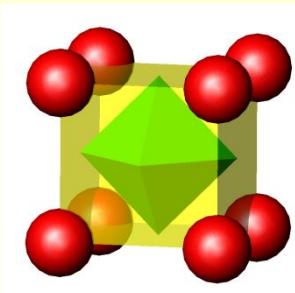
▪ Materials

- $\text{La}_{0.58}\text{Sr}_{0.4}\text{Co}_{0.2}\text{Fe}_{0.8}\text{O}_{3-\delta}$ (LSCF)
- $\text{La}_{0.6}\text{Sr}_{0.4}\text{CoO}_{3-\delta}$ (LSC)
- $\text{Nd}_2\text{NiO}_{4+\delta}$ (NNO)

SOFC cathodes

Perovskites ($\text{ABO}_{3-\delta}$):

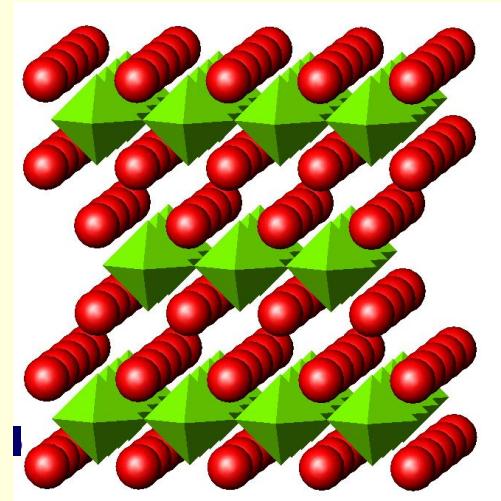
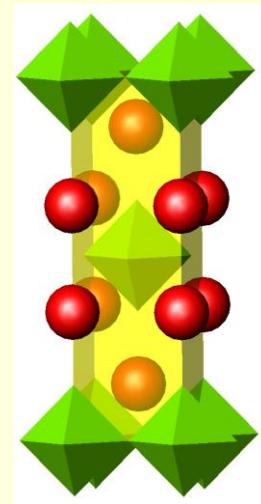
A=La,Sr,Ba and B=Co,Fe,Mn
 δ ... oxygen nonstoichiometry
(oxygen vacancies)



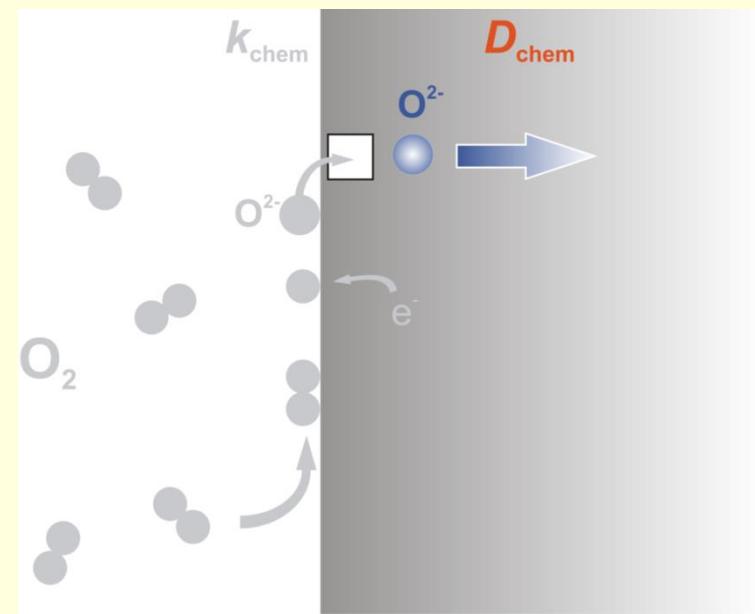
A-site ions (red)
 BO_6 -octahedra (green)

K_2NiF_4 -type oxides ($\text{A}_2\text{BO}_{4+\delta}$):

A=Nd,Pr and B=Ni
 δ ... oxygen nonstoichiometry
oxygen excess (oxygen interstitials)



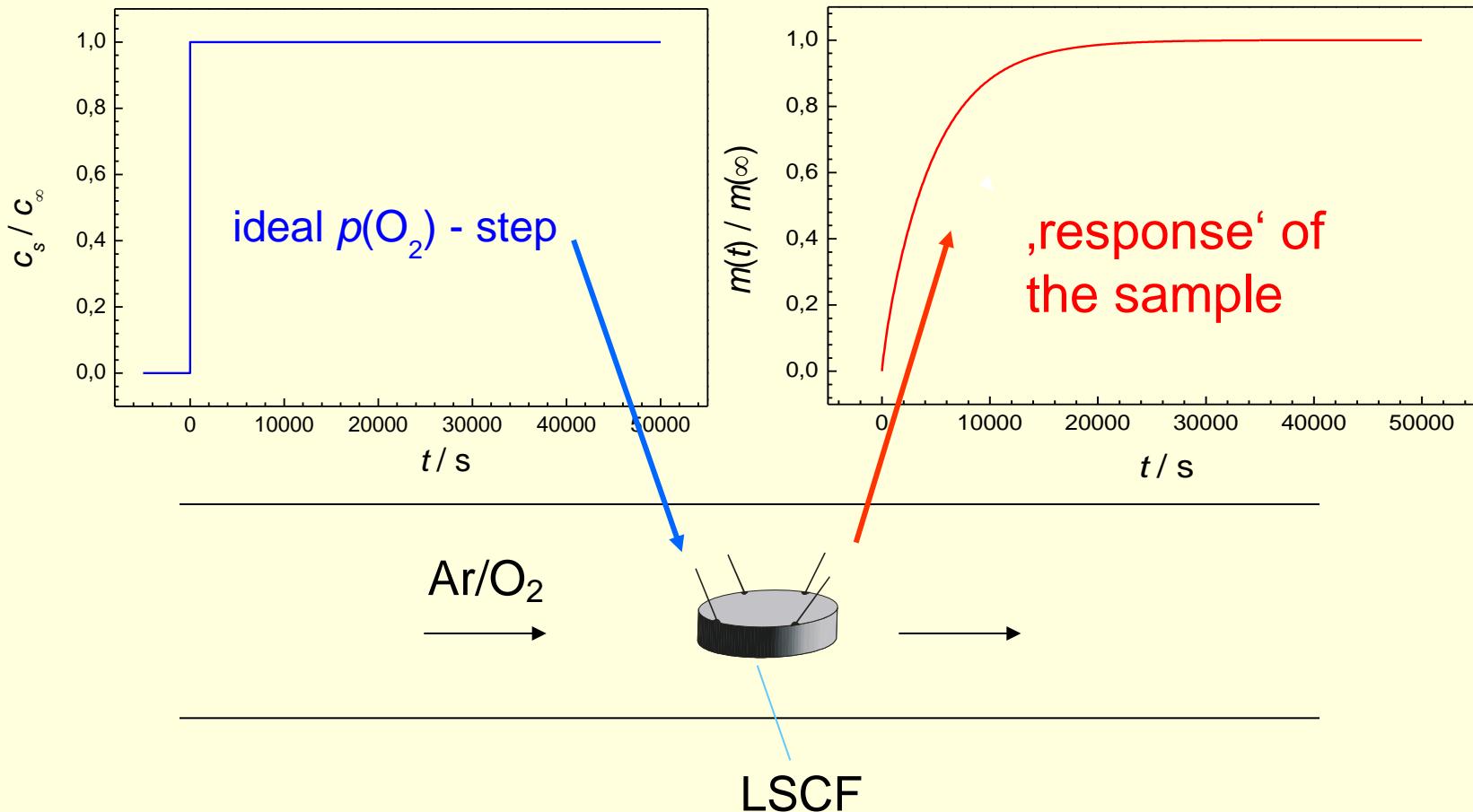
Oxygen exchange kinetics



- oxygen adsorption on surface
 - physisorption: $O_2(g) \rightarrow O_{2,ads}$
 - chemisorption: $O_2(g) \xrightarrow{e^-} O_{2,ads}^-$ or $O_2(g) \xrightarrow{2e^-} O_{2,ads}^{2-}$
- oxygen dissociation on surface (involving oxygen vacancies)
- incorporation of oxygen into the lattice $O_{ads}^- + V_O^{\bullet\bullet} + e' \rightarrow O_O^x$

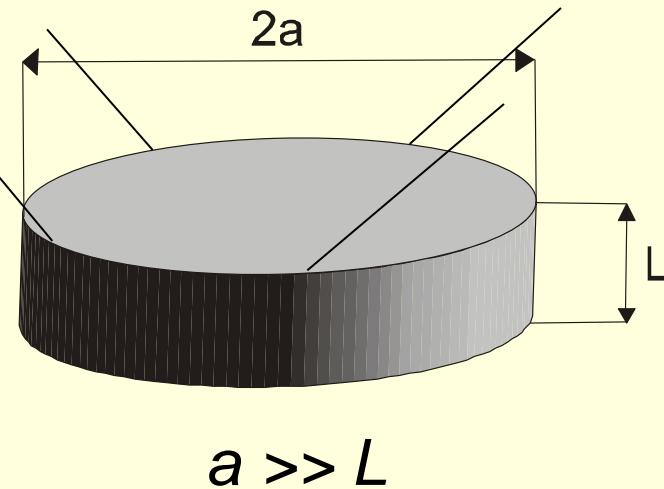
Oxygen exchange kinetics

- dc-conductivity relaxation technique



Oxygen exchange kinetics

- dc-conductivity relaxation technique



$$\frac{\partial c}{\partial t} = \tilde{D} \frac{\partial^2 c}{\partial x^2}$$

$$\tilde{D} \frac{\partial c}{\partial x} = \pm \tilde{\kappa} (c - c_s) \quad x = 0, L$$

\tilde{D} ... chemical diffusion coefficient

$\tilde{\kappa}$... surface exchange coefficient

$$\frac{\sigma(t) - \sigma(0)}{\sigma(\infty) - \sigma(0)} = \frac{m(t)}{m(\infty)}$$

van der Pauw method $\rightarrow \sigma$

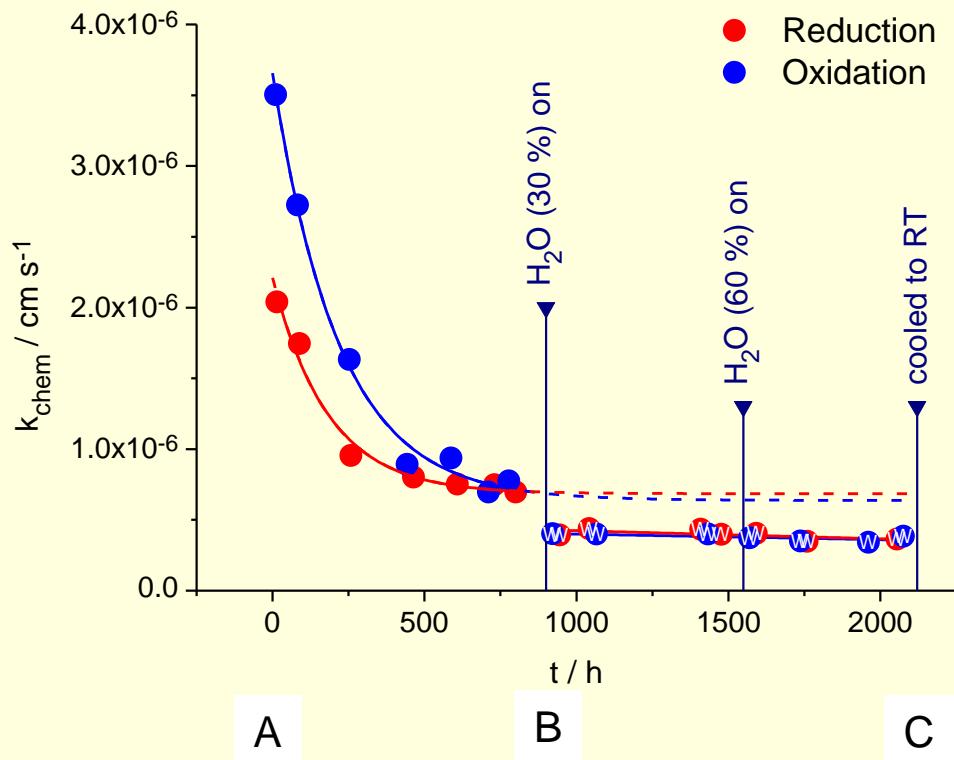
Long-term stability of the oxygen surface exchange of

- $\text{La}_{0.58}\text{Sr}_{0.4}\text{Co}_{0.2}\text{Fe}_{0.8}\text{O}_{3-\delta}$
- $\text{Nd}_2\text{NiO}_{4+\delta}$

in dry and humid O_2/Ar -atmospheres

$\text{La}_{0.58}\text{Sr}_{0.4}\text{Co}_{0.2}\text{Fe}_{0.8}\text{O}_{3-\delta}$

- Long term oxygen exchange kinetics of $\text{La}_{0.58}\text{Sr}_{0.4}\text{Co}_{0.2}\text{Fe}_{0.8}\text{O}_{3-\delta}$ in dry and wet atmospheres at 600°C



$\text{La}_{0.58}\text{Sr}_{0.4}\text{Co}_{0.2}\text{Fe}_{0.8}\text{O}_{3-\delta}$
elemental depth
profiles

fresh sample (A)

La, Sr, Co, Fe

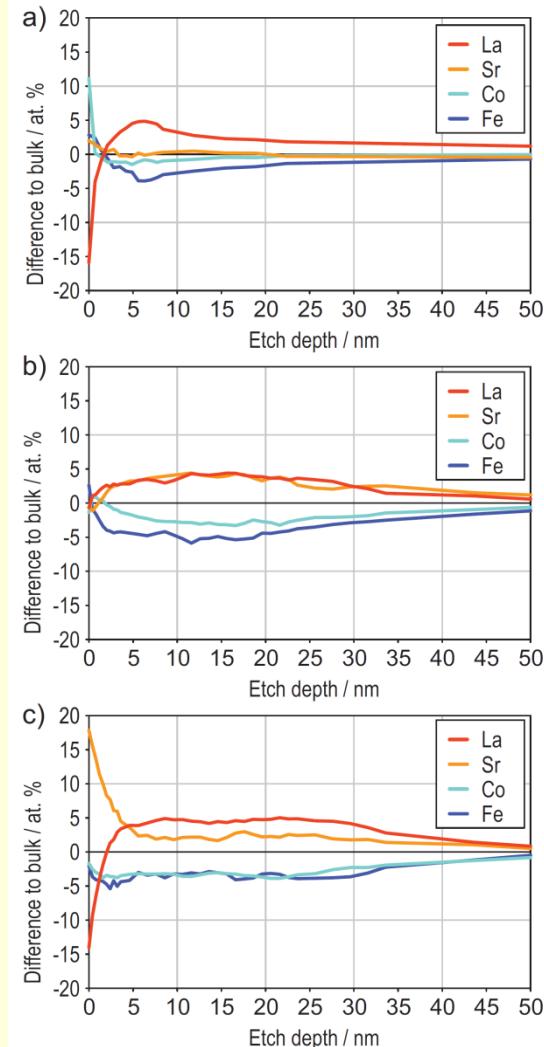
+ O and Si

1000h
dry 1% O_2/Ar
atmosphere (B)

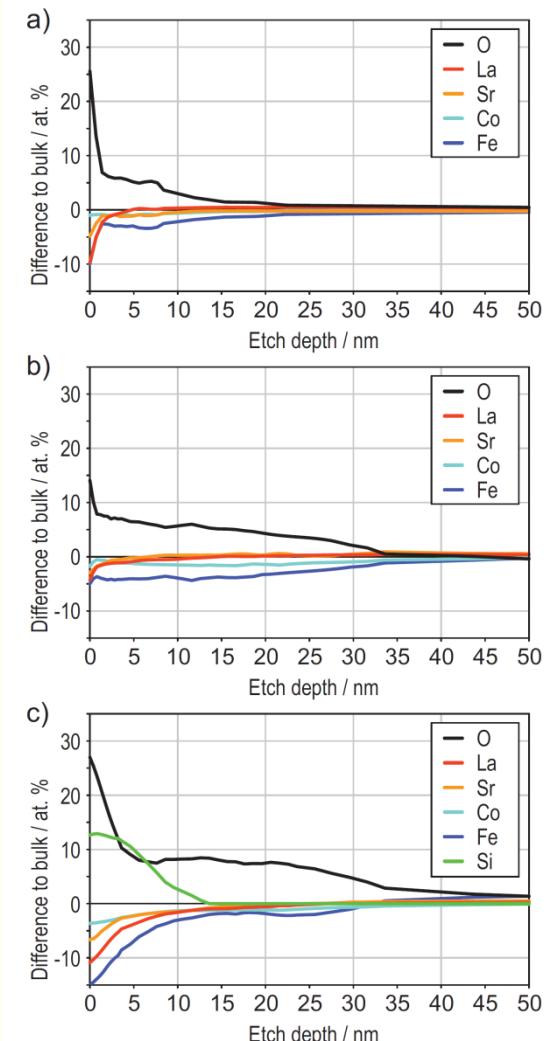
1000h dry + 1000h
humid 1% O_2/Ar -
atmosphere (C)

E. Bucher et al, Solid State
Ionics, 191 (2011) 61-67

Eco-Mobility 2014, Vienna



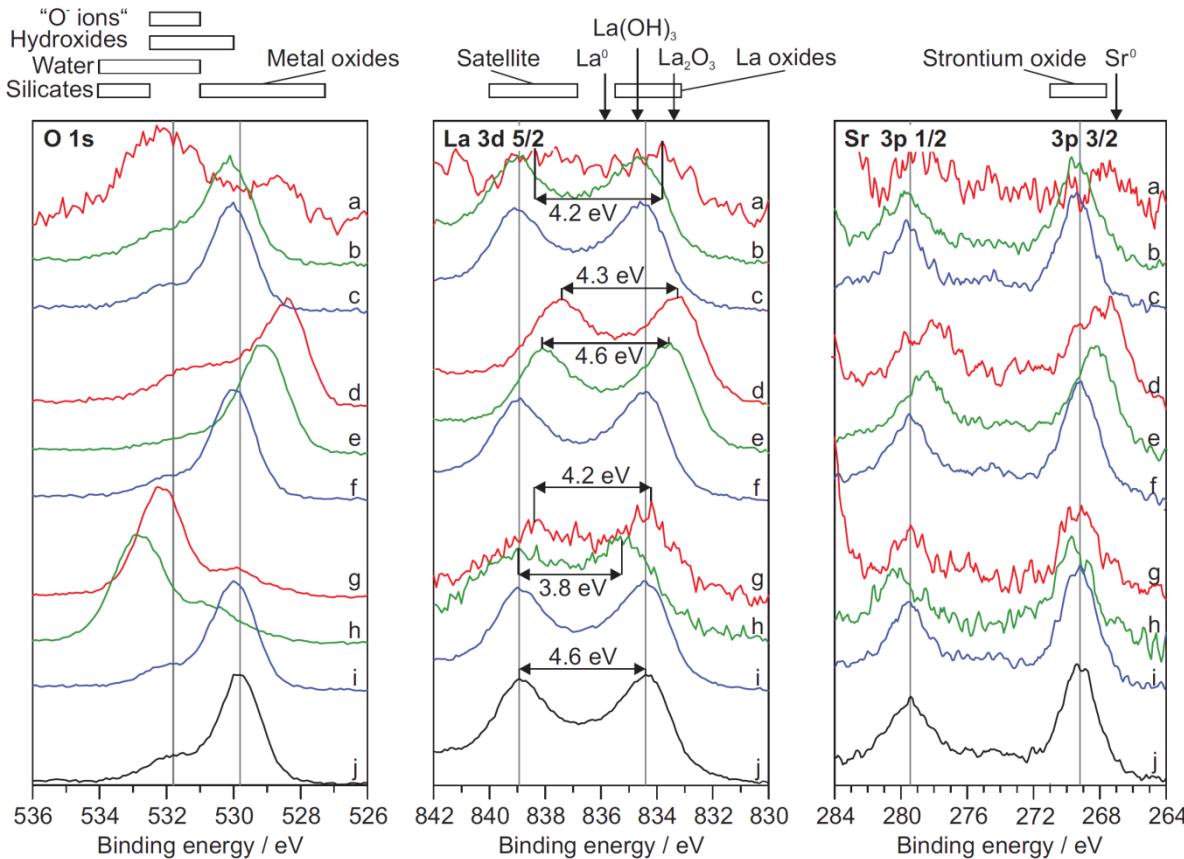
constant cation composition : ~ 50 nm.



$\text{La}_{0.58}\text{Sr}_{0.4}\text{Co}_{0.2}\text{Fe}_{0.8}\text{O}_{3-\delta}$

core level spectra

O 1s, La 3d, Sr 3p



As prepared sample

- a) 0 nm
- b) 1 nm
- c) 10 nm

After 1000 h in a dry atmosphere

- d) 0 nm
- e) 1 nm
- f) 10 nm

After additional 1000 h in a humid atmosphere

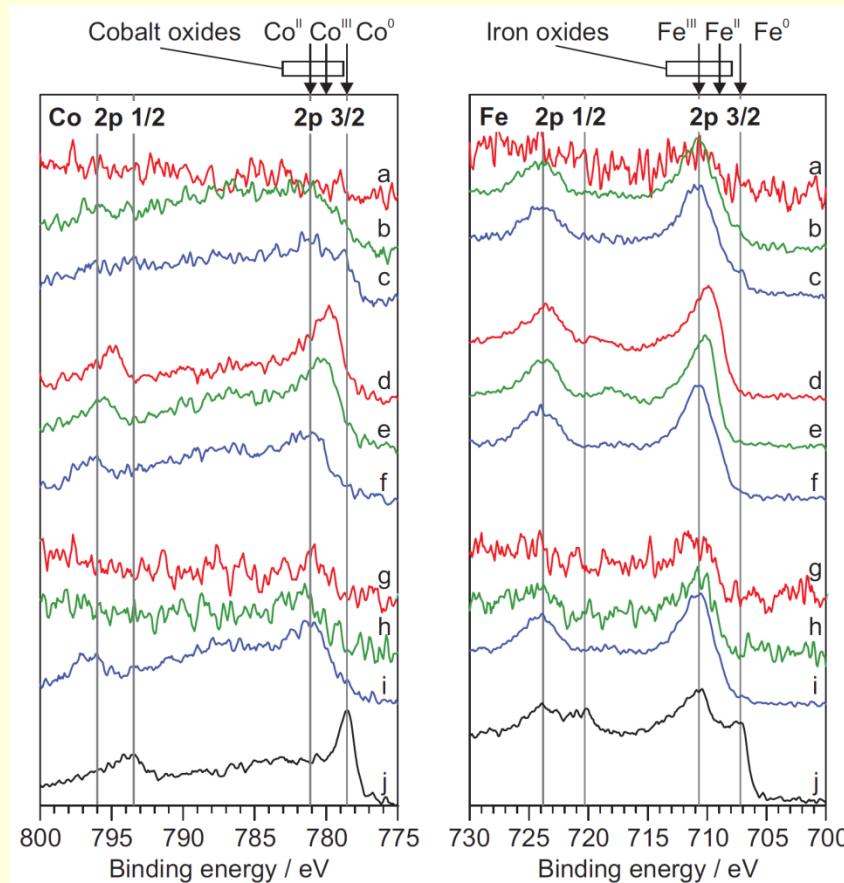
- g) 0 nm
- h) 1 nm
- i) 10 nm

Bulk - reference
j) > 300 nm

XPS core level spectra of the O 1s, La 3d, Sr 3p peaks of the three different samples, obtained from immediate surface (0 nm), at etch depths of 1 nm and 10 nm, and from bulk (depth > 300 nm).

$\text{La}_{0.58}\text{Sr}_{0.4}\text{Co}_{0.2}\text{Fe}_{0.8}\text{O}_{3-\delta}$ core level spectra

Co 2p, Fe 2p



As prepared sample

- a) 0 nm
- b) 1 nm
- c) 10 nm

After 1000 h in a dry atmosphere

- d) 0 nm
- e) 1 nm
- f) 10 nm

After additional 1000 h in a humid atmosphere

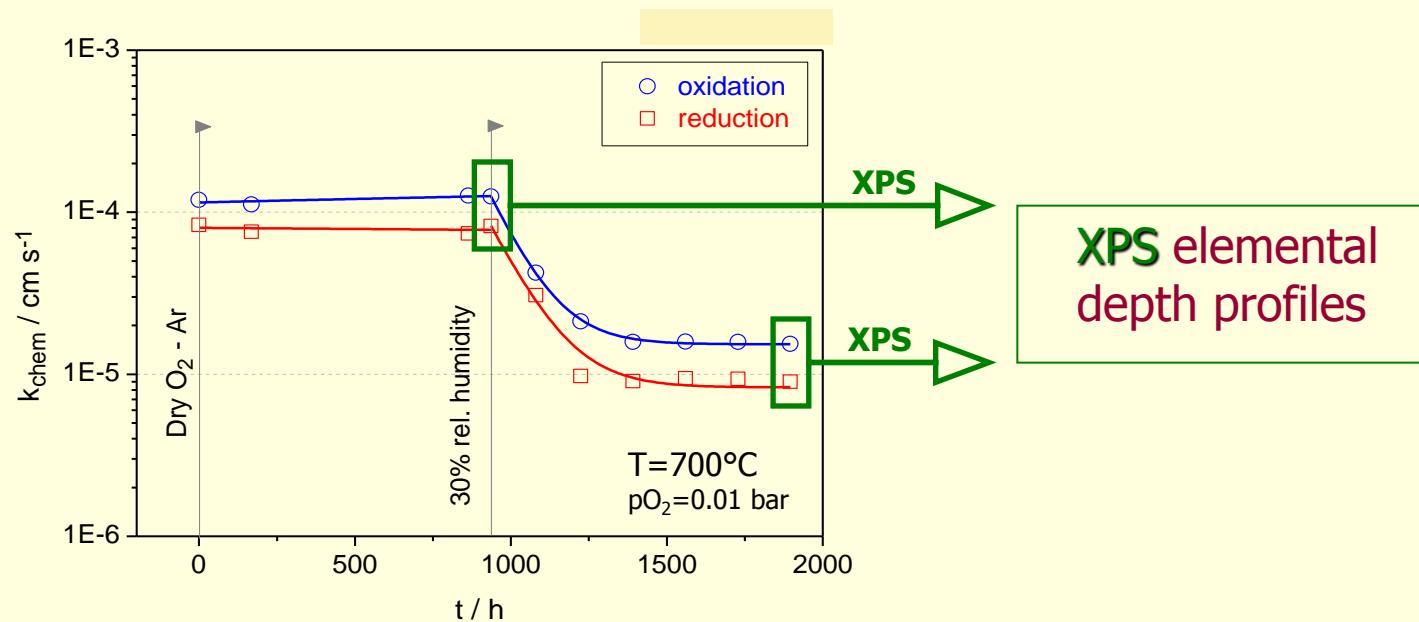
- g) 0 nm
- h) 1 nm
- i) 10 nm

Bulk - reference
j) > 300 nm

XPS core level spectra of Co 2p and Fe 2p peaks of the three different samples, obtained from immediate surface (0 nm), at etch depths of 1 nm and 10 nm, and from bulk (depth > 300 nm).

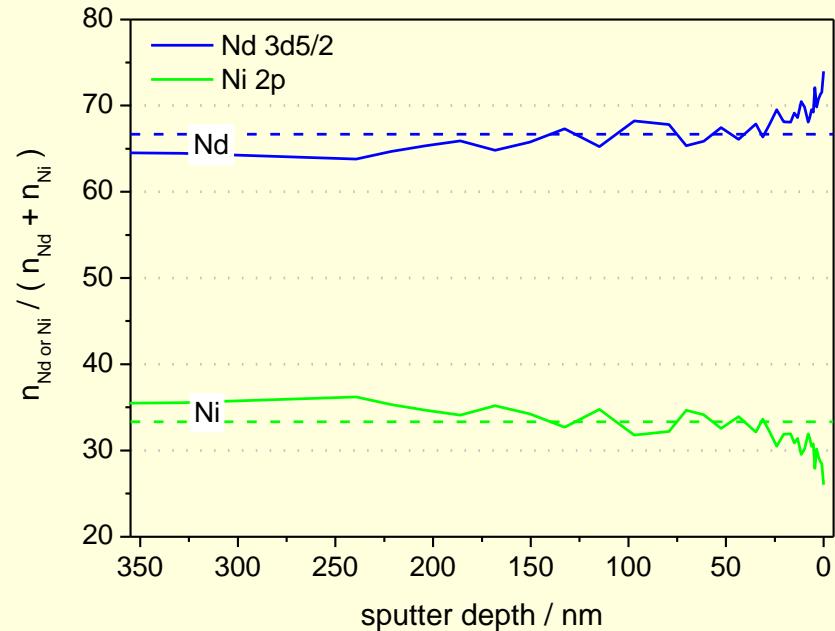
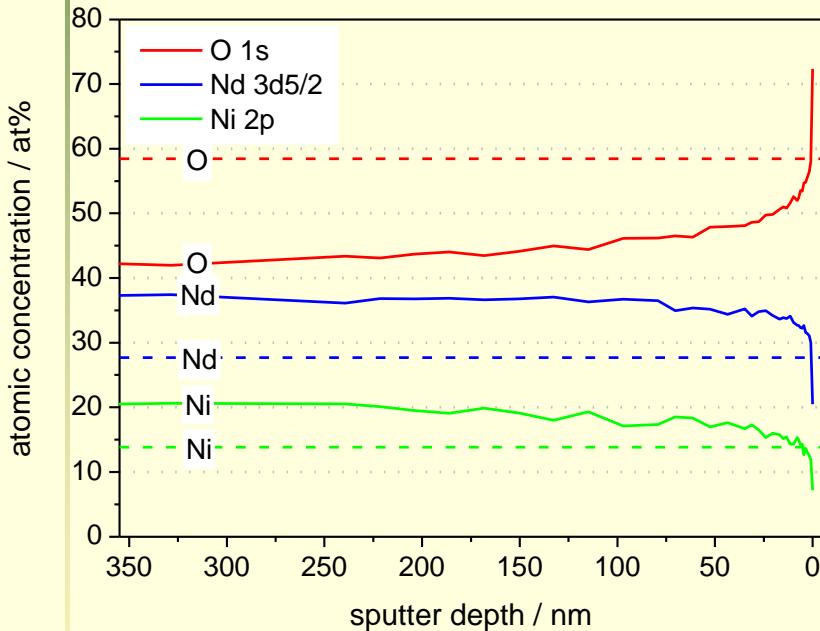
Nd₂NiO_{4+δ}

- Long term stability of k_{chem} of Nd₂NiO_{4+δ} in dry + wet atmospheres (700°C)



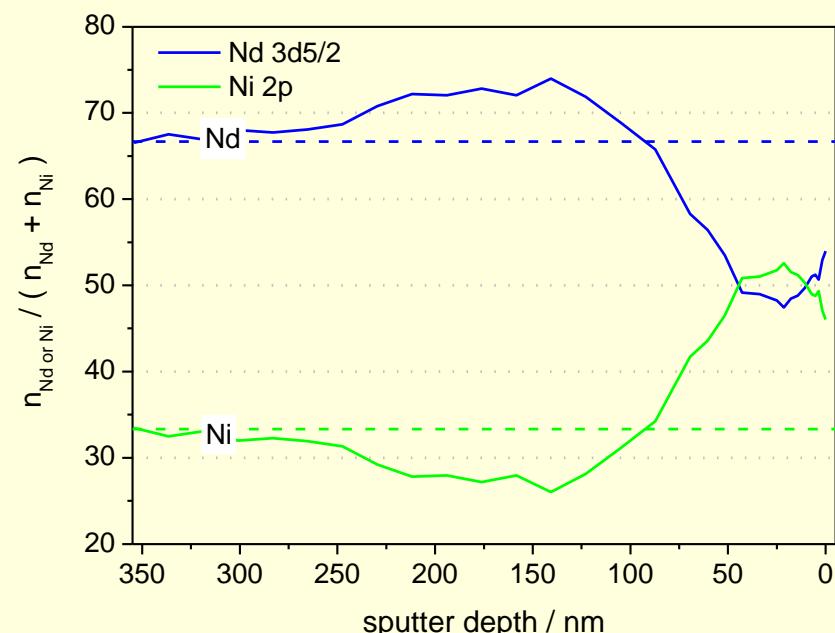
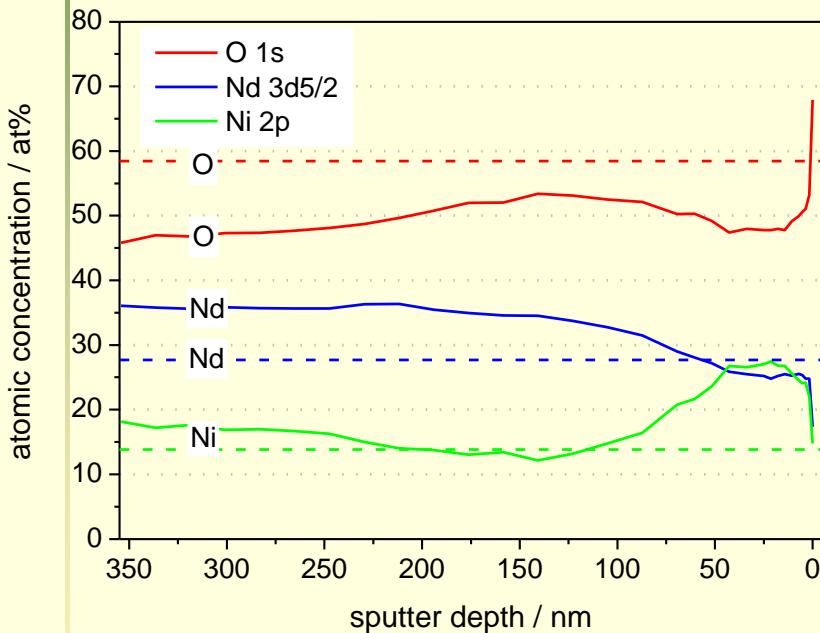
Nd₂NiO_{4+δ} : XPS elemental depth profiles

Nd₂NiO_{4+δ} 1000h dry atmosphere
T = 700° C pO₂ = 0.01 bar



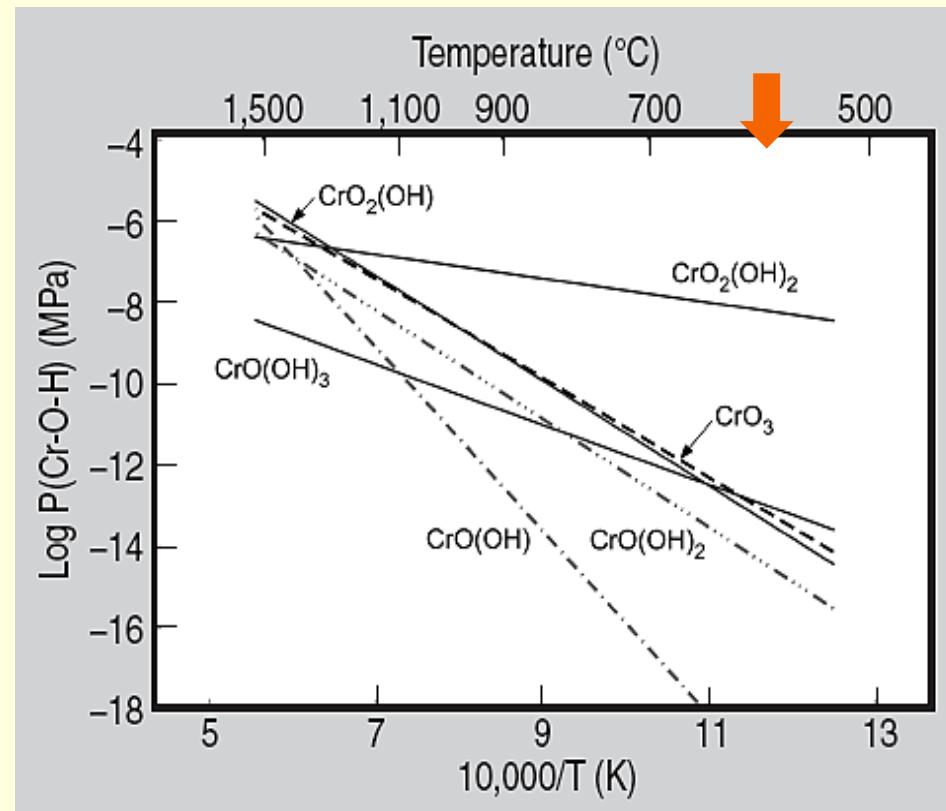
Nd₂NiO_{4+δ} : XPS elemental depth profiles

Nd₂NiO_{4+δ} 1000h dry + 1000h humid atmosphere
T = 700° C pO₂ = 0.01 bar



Gas phase equilibria

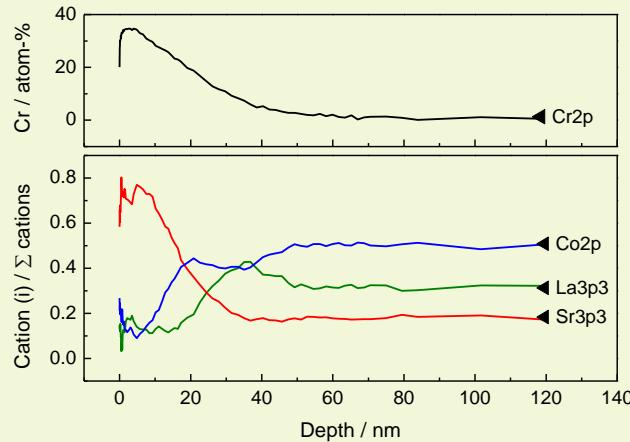
- ▶ Poisoning (degradation) is caused by reaction of the cathode with **volatile species** which are **transported via the gas phase**
- ▶ The predominating gas phase species depend on T, $p(O_2)$, $p(H_2O)$
- ▶ Under the present experimental conditions:
 - $\frac{1}{2} Cr_2O_3(s) + H_2O(g) + \frac{3}{4} O_2(g) \rightarrow CrO_2(OH)_2(g)$
 - $\frac{1}{2} Cr_2O_3(s) + \frac{3}{4} O_2(g) \rightarrow CrO_3(g)$
 - $SiO_2(s) + 2 H_2O(g) \rightarrow Si(OH)_4(g)$



E. Opila et al., JOM 58 (2006) 22.

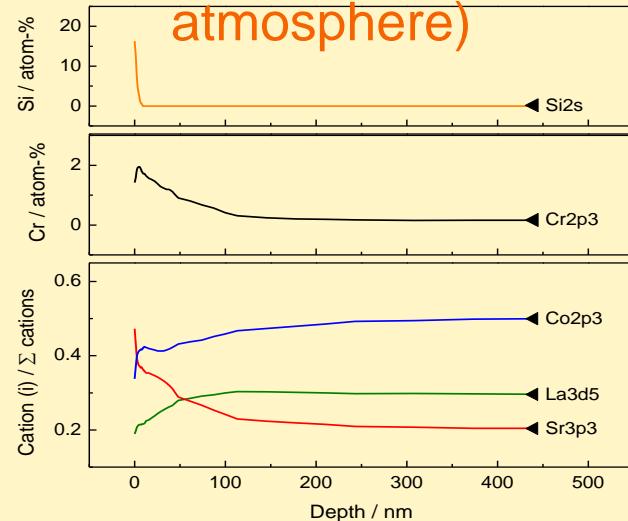
Comparison: Cr-layer vs. Cr-pellet

$(\text{La}, \text{Sr})\text{CoO}_{3-\delta}$ 1000 h 600°C
 Cr-layer (dry atmosphere)



1. Cr present up to about 40 nm depth
2. **Sr-enrichment** up to 40 nm depth
3. Applied Cr-layer → higher Cr-content (<35 at-%)
4. Cr-content in 30 nm+90 nm depth: **0.5 at-%**
5. No Si detected (dry atmosphere!)

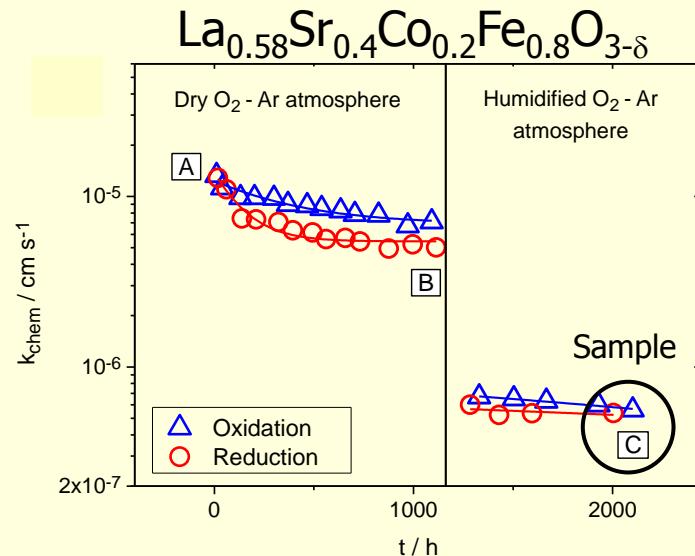
... with Cr-pellet (humid atmosphere)



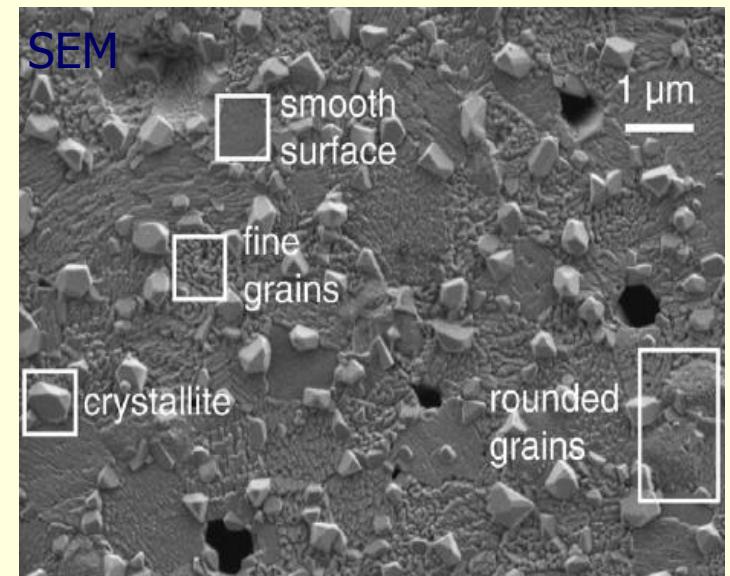
1. Cr present up to about 100 nm depth
2. **Sr-enrichment** up to 100 nm depth
3. External Cr-source → lower Cr-content (<2 at-%)
4. Cr-content in **90 nm depth: 0.5 at-%**
5. Thin zone of Si-poisoning (wet atmosphere!)

Scanning electron microscopy

k_{chem} in dry and humid atmosphere

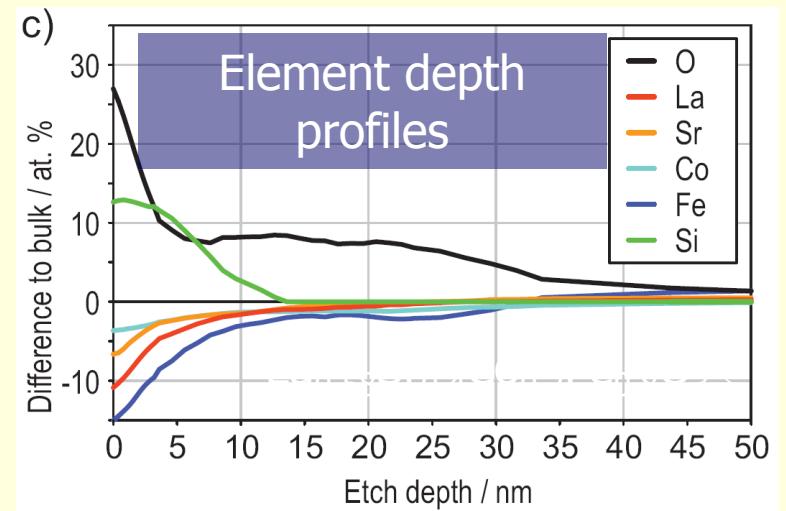
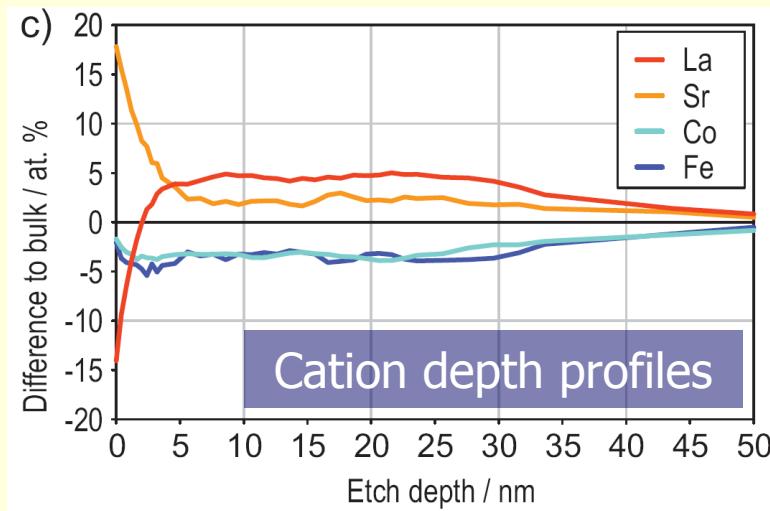


Information on surface topography (sample C)

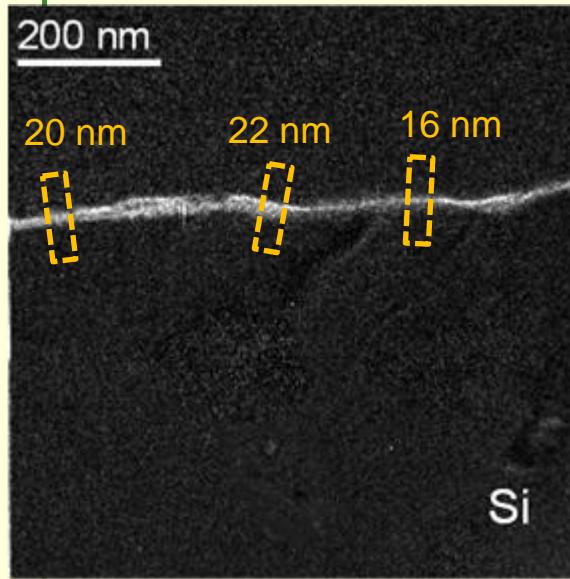


X-ray photoelectron spectroscopy

XPS → Average chemical composition of surface in the analyzed zone (around 500 µm)
after 1000h dry + 1000h humid 1% O₂/Ar-atmosphere

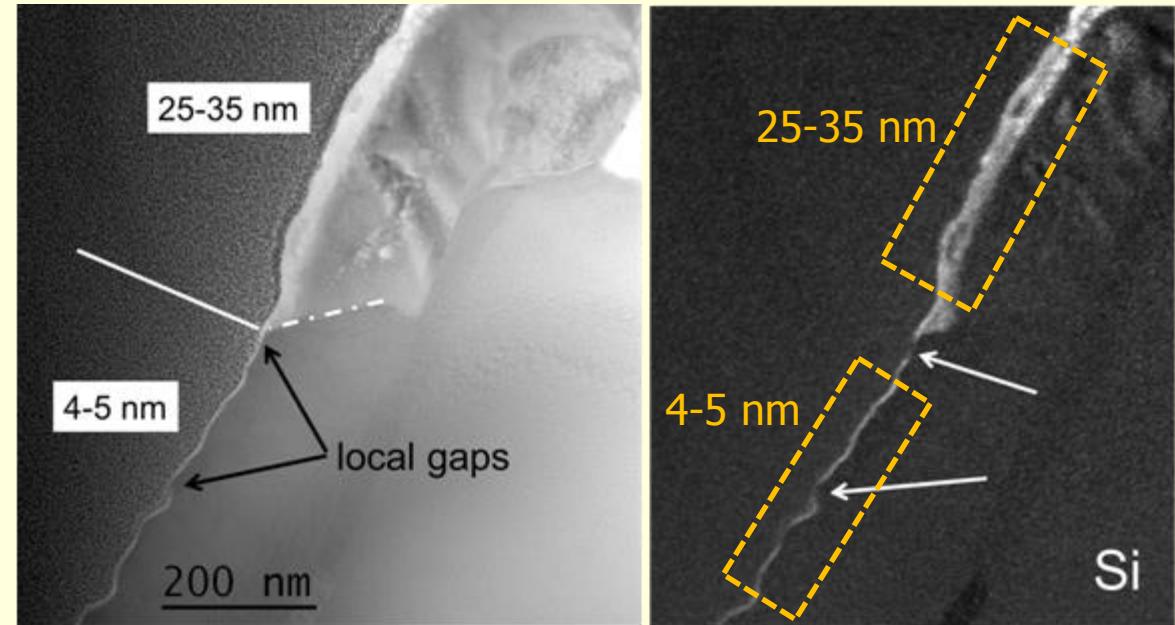


Silicon poisoning (EFTEM-SI)



- Energy-filtered TEM spectrum-imaging
- Si is present throughout the surface
- Thickness of the layer approx. 5-35 nm

- Thinner and thicker regions coexist



E. Bucher, C. Gspan, F. Hofer, W. Sitte, Solid State Ionics, 230 (2013) 7–11

Ag activated oxygen exchange of $\text{La}_2\text{NiO}_{4+\delta}$

Ag-activated oxygen exchange of $\text{La}_2\text{NiO}_{4+\delta}$

■ Problem

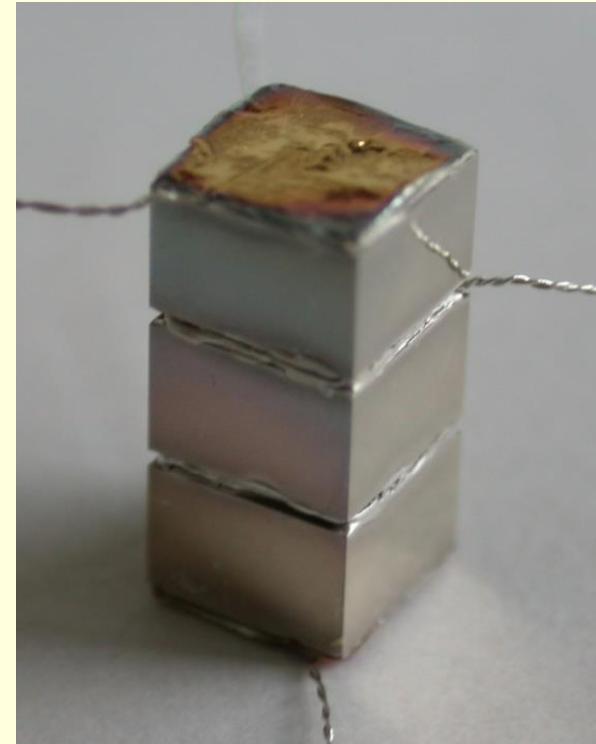
- Surface limited oxygen exchange of $\text{La}_2\text{NiO}_{4+\delta}$
- even large samples 6 x 6 mm do not allow the determination of D_{chem}



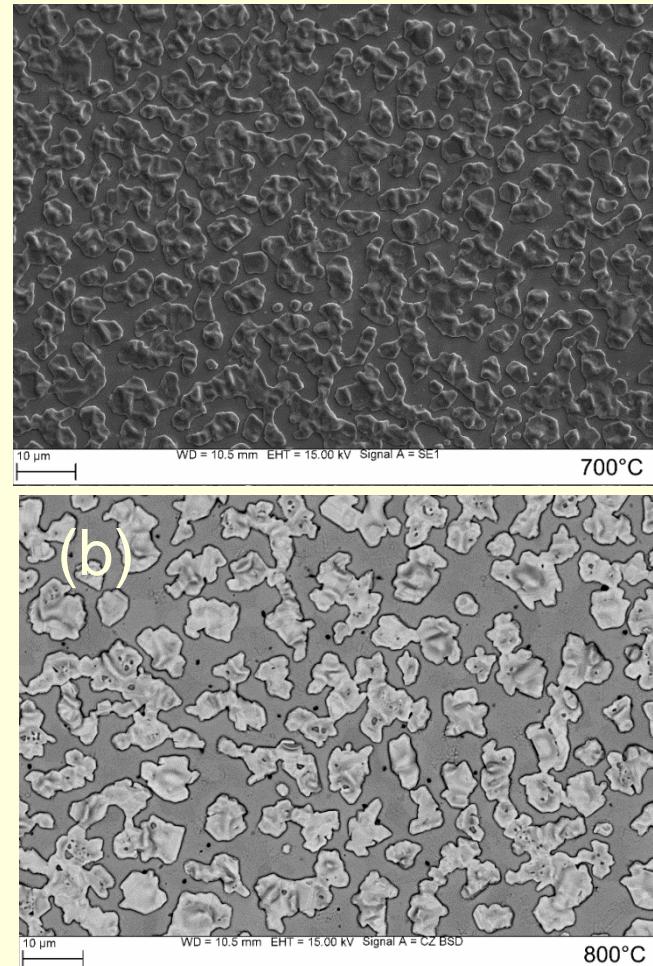
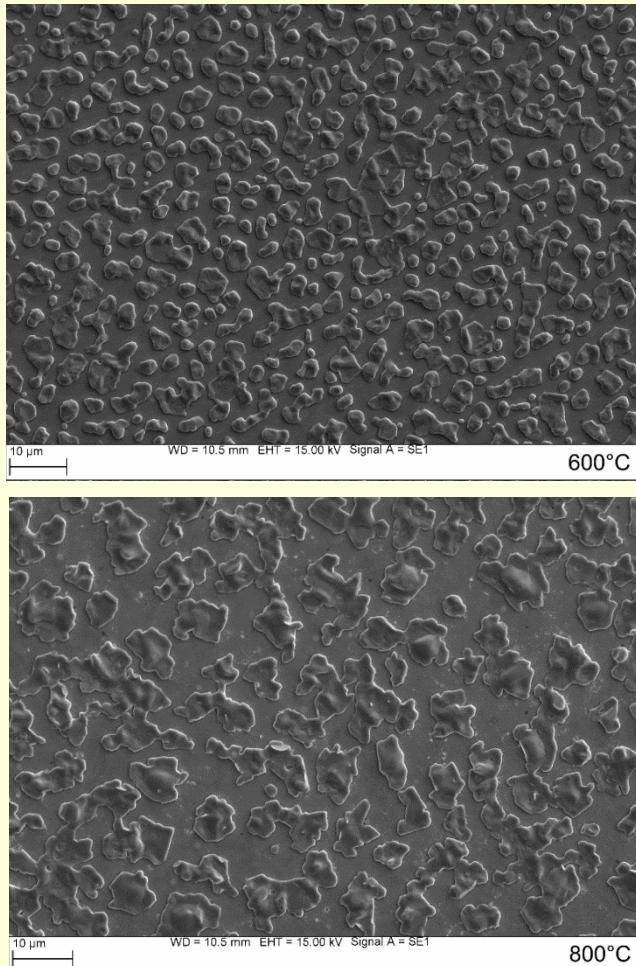
Ag-activated oxygen exchange of $\text{La}_2\text{NiO}_{4+\delta}$

■ Solution

- $\text{La}_2\text{NiO}_{4+\delta}$ is covered by a 200 nm-thick Ag-layer
- gold plated at the ends
- enhanced oxygen exchange:
 D_{chem} can be determined with usual sample sizes

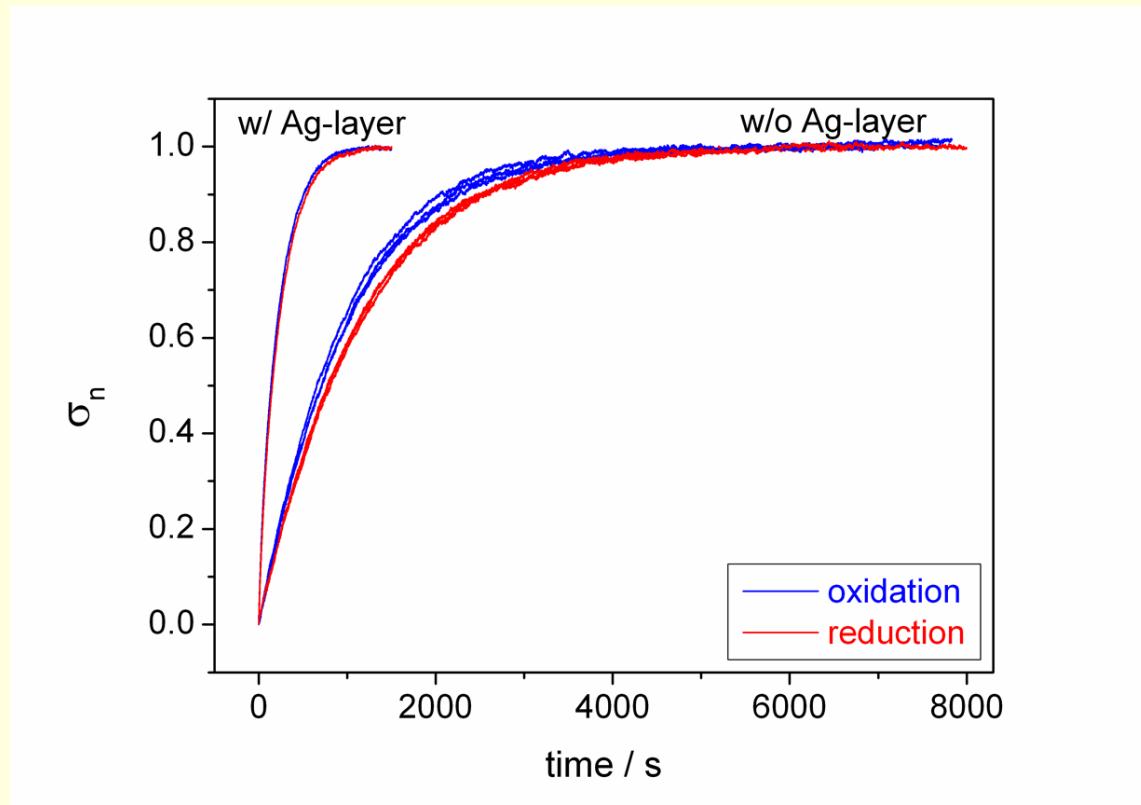


Ag-activated oxygen exchange of $\text{La}_2\text{NiO}_{4+\delta}$



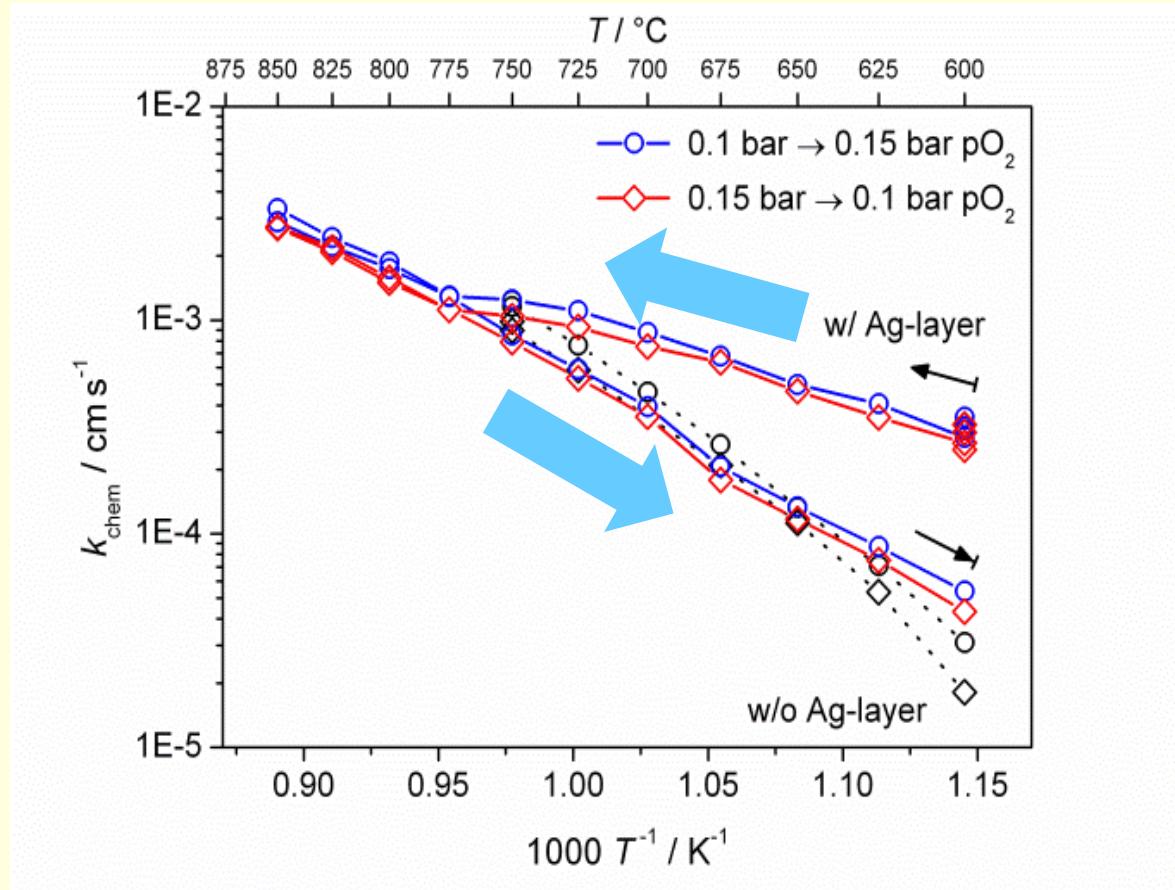
(a) SEM-pictures of a silver film on $\text{La}_2\text{NiO}_{4+\delta}$ after annealing for 24 hours at 600, 700 and 800°C. (b) Backscatter image

Ag-activated oxygen exchange of $\text{La}_2\text{NiO}_{4+\delta}$



Conductivity relaxation curves of $\text{La}_2\text{NiO}_{4+\delta}$ with and without a surface layer of silver.

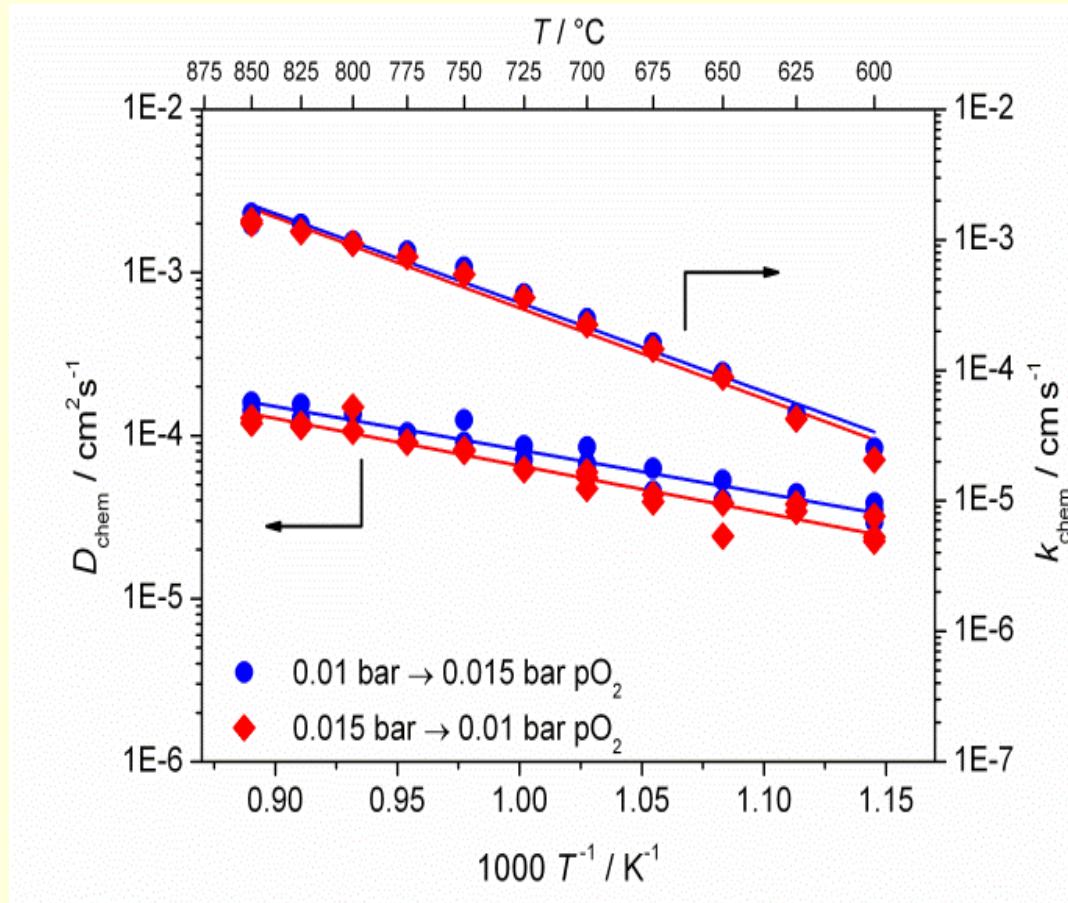
Ag-activated oxygen exchange of $\text{La}_2\text{NiO}_{4+\delta}$



Effect of a 200nm Ag layer on k_{chem} of $\text{La}_2\text{NiO}_{4+\delta}$ at $p\text{O}_2 = 0.1$ bar.

A. Egger, W. Sitte, Solid State Ionics 258 (2014) 30–37

Ag-activated oxygen exchange of $\text{La}_2\text{NiO}_{4+\delta}$



Arrhenius plot of D_{chem} and k_{chem} of oxygen for $\text{La}_2\text{NiO}_{4+\delta}$ at $p\text{O}_2 = 0.01$ bar.

Conclusions

- **Long time investigations (1-2 kh)** → necessary to investigate significant changes of the oxygen exchange kinetics of **complex oxides** in dry or humid O₂/Ar-atmospheres.
- k_{chem} of **Sr-containing SOFC cathodes** → more severely affected in dry and humid atmospheres than Sr-free alternatives.
- **XPS and elemental depth profiles, HRTEM investigations** → insight into the origin of the degradation in the range 0-50 nm.
- **Humid atmospheres** → strongly enhance transport of Si and Cr via the gas phase in the presence of a Si- or Cr-source, even at temperatures as low as 600°C.
- **Surface limited oxygen exchange:** increase of k_{chem} by deposition of a thin layer of catalytically active nanocrystalline Ag.